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1 **A machine learning based multi-scale computation framework for**

2 **granular materials**

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10 **Abstract**

11 With the development of experimental measurement technology and high-fidelity numerical
12 simulations of granular materials, empirical-based classical constitutive models may not be able to
13 take full advantage of the rapidly increasing available datasets. Machine learning-based models
14 can inherently avoid phenomenological assumptions to directly learn the constitutive relationship
15 from the datasets, and the trained model is sufficiently flexible to be reconstructed once new
16 training samples are added. In this work, a coupled finite element method and machine learning
17 (FEM-ML) computational framework is proposed for simulating granular materials. Gaussian
18 process-based random loading paths and coupled FEM-DEM simulations are used to generate
19 training samples. A parametrisation of the material deformation history is used to represent the
20 historical influence of granular materials. An uncertainty-level based active learning is utilised to
21 evaluate the informativeness of data points for network training and then to establish an effective
22 resampling scheme from a massive dataset. Two examples are provided to show the applicability
23 of the implemented FEM-ML framework. The performance of the proposed framework is also
24 evaluated, the error is systematically analysed, and possible improvements are discussed. The
25 results demonstrate that the FEM-ML framework offers considerable improvements in terms of
26 computational efficiency and competence to simulate the mechanical responses of granular
27 materials.

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28 **1 Introduction**

29 Granular materials such as gravel, sand, and powders are ubiquitous in industry and geotechnical
30 applications, and they are considered to be the second most abundant material on Earth after fluids
31 [1]. The discrete nature of granular materials and dissipative interactions among particles give rise
32 to a rich and complex bulk behaviour, making them differ significantly from solids, liquids, and
33 gases. The complexity of granular media can be partially attributed to its unique features, such as
34 inherent anisotropy and heterogeneity [2, 3], pressure and rate-dependence [4–6], continuous
35 evolving microstructure and complicated strain localisation phenomenon within unstable granular
36 materials [7–11]. Accurately reproducing the mechanical behaviour of granular materials subject
37 to various external loads through mathematical equations is intricate but crucial for engineering-
38 scale numerical simulations.

39

40 In general, the mechanical behaviour of history-dependent granular materials has been described
41 by empirical constitutive laws formulated within continuum thermodynamics and elastoplastic
42 theory. Internal variables characterise the material state according to phenomenological
43 assumptions. Advancements in **micro-mechanical** simulations and the internal fabric statistical
44 description [12–15] have inspired physics-based internal variables that encapsulate the micro-
45 structure and motivated the bottom-up mathematical modelling of granular materials by linking
46 contact fabric tensor to macro-mechanical responses [6, 15–18]. Nearly in parallel, another class
47 of bottom-up modelling approaches, the hierarchical **multi-scale** modelling, which directly bridges
48 the macroscopic response of granular materials and the macroscopic boundary conditions [19–24],
49 has also been developed due to computer hardware development. In the framework of hierarchical
50 **multi-scale** modelling, the models of two scales are solved concurrently, and the results are
51 exchanged on-the-fly. Both the two conventional modelling paradigms have achieved quite a
52 success.

53

54 However, there are still considerable challenges in the way of their further progress. For the
55 mathematical-based phenomenological approach, uncertainty arises due to the imperfect knowledge of
56 the functional form of the constitutive laws and the parameter calibration is subject to the

57 increasing complexity. On the other hand, the classical hierarchical **multi-scale** computation is
58 indecently time-consuming because a large number of lower-scale simulations are concurrently
59 required during the computation of a macroscopic model.

60

61 In addition to the continuum mechanics modelling and hierarchical **multi-scale** modelling, the use
62 of neural-network-based constitutive models can be traced back to the 1990s and the beginning of
63 the 21st century [25–29]. Data-science-based model is a general term used for the recently arising
64 new modelling method related to massive datasets, statistics, machine learning and data mining.
65 **Some related work in the fields of computational mechanics, geomechanics and geotechnical**
66 **engineering can be found in recent reviews [30–34] and references therein.**

67

68 In the case of the surrogate model constructed by the neural network, [27, 35–41], material
69 informatics uses database techniques to identify parameter correlations and then uses machine-
70 learning regression techniques to provide the quantitative predictive model. Due to the powerful
71 high-dimensional mapping capability of the neural network, this method can reproduce the
72 constitutive response of materials more precisely than the traditional mathematical based models,
73 provided that sufficient data is available. For hierarchical **multi-scale** computation, a promising
74 application of the neural network-based approach is to accelerate **multi-scale** mechanical
75 calculations because of the higher computational efficiency over the conventional framework [37,
76 42, 43].

77

78 Unlike the regression-based modelling approach, another kind of data-driven method directly
79 chooses the samples which best satisfy the compatibility and equilibrium constraints from the
80 prepared database composed of strain stress pairs [44–47]. Instead of calibrating the parameters of
81 the pre-constructed model through the datasets, this pure data-based method, which is solely based
82 on the datasets, transforms the solution problem into a minimum distance optimisation, further
83 avoiding the errors associated with the regression process, and almost wholly reproduces the
84 results of the conventional multi-scale method. However, the particularity and computational
85 efficiency of this *ad hoc* method may hinder its further development.

86

87 History dependence, fundamentally the intrinsic characteristic of granular materials [48], is
88 another challenge for the data-driven modelling of granular materials. Internal variables based on
89 plasticity mechanics approaches are often used to describe the historical state of the material in
90 traditional elastoplastic constitutive models.

91

92 In the data-driven modelling work, some scholars use RNN (recurrent neural networks) to deal
93 with the material history dependence and their works have demonstrated that stress-strain
94 responses of granular materials can be predicted satisfactorily in certain stress-strain spaces [37,
95 38, 49–52]. However, the training is normally quite time-consuming as this model is not suitable
96 for GPU parallel training because of its inherent sequential nature. Additionally, using different
97 step sizes not included in the training samples will deteriorate the computational stability once the
98 model is employed in boundary value problems [49, 50]. Due to the development of granular
99 material simulation and the emergence of internal fabric description methods, some scholars use
100 energy-based variables or fabric tensors to calibrate the current state of granular materials [46, 47].
101 This approach requires encoding the loading history of the material as the describer of the current
102 state with as few parameters as possible, but how to take the sequence of macroscopic information
103 (e.g. strain, stress, etc.) and encode it into historical state variables is again a tricky problem. Also,
104 some preceding work attempts to discover some parameters to encode the history strain-stress path
105 experienced by materials. In one example [39], the accumulation of the absolute values of the
106 strain increments is used to uniquely represent the history state and appears to perform well in
107 boundary value problem simulations.

108

109 The tangent operator of the strain-stress relation is necessary for the nonlinear iteration using the
110 Newton-Raphson method in FEM simulations. In contrast to the FEM simulation of continuous
111 materials, the particle simulation of granular materials, such as DEM (discrete element method),
112 exhibits a random nature. The stress-strain curve can rise and drop sharply after entering the
113 plastic regime, so the Jacobians of the network computed stress to the input strain cannot represent
114 a proper tangent operator of the material. **The approach that uses the perturbation method to**
115 **numerically compute the Jacobian performs poorly in nonlinear iteration [22]. Therefore, the**
116 **partial derivative tangent operator [28, 37] used for continuous materials may not work well for**

117 granular materials. A viable solution, to be introduced in Section 3, is to replace the tangent matrix
118 with a secant matrix in the non-linear iteration.

119

120 This work aims to propose a neural-network-based constitutive model as the surrogate model that
121 directly learns the multi-scale mechanical relationship from the raw data collected from the
122 consistent FEM-DEM multi-scale simulation. This model can be integrated into the FEM program
123 with minimal intrusion and is considerably more computationally efficient than the classical multi-
124 scale procedure. The history effect of the strain-stress relationship is considered by an internal
125 variable-based parameterisation. The secant matrix is adopted to replace the Jacobian operator in
126 the nonlinear iteration.

127

128 The rest of the paper is organised as follows. We first introduce the FEM-DEM framework in
129 Section 2, followed by the description of the neural network-based constitutive model, a self-
130 consistent sampling method, a general procedure of the uncertainty level-based active learning
131 method and the model training process in Section 3. We first present the baseline numerical
132 simulation of a biaxial compression problem to validate our implemented framework in Section 4.
133 This is subsequently employed in a retaining wall simulation to evaluate its generalisability. The
134 advantages and disadvantages of the FEM-ML framework and possible further improvements are
135 summarised in the conclusion.

136

137 **2 Multi-scale FEM-DEM framework**

138 The multi-scale FEM-DEM framework for solving a granular problem consists of two main parts:
139 (1) macro model which is implemented in FEM; (2) representative volume element (RVE) which
140 is concurrently simulated through DEM. The main procedures in the FEM-DEM framework are
141 shown in Figure 1.

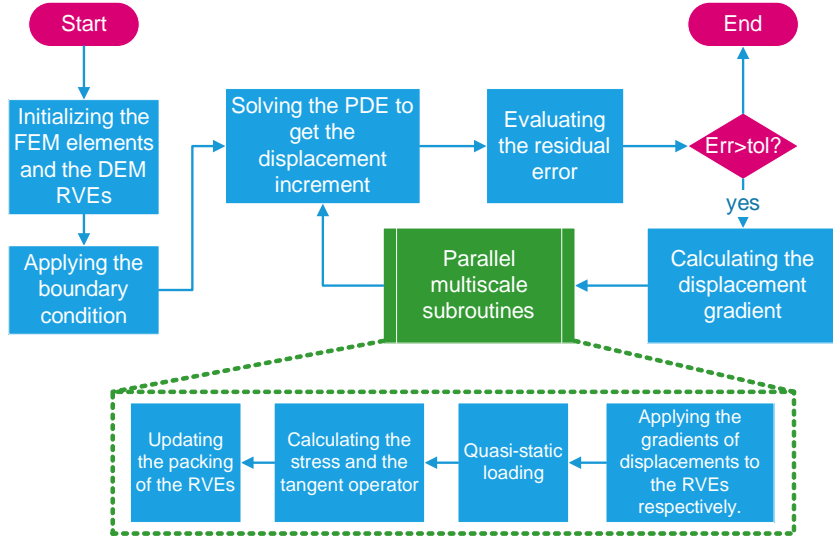


Figure 1 Flow chart of the FEM-DEM multiscale scheme

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145 2.1 Macro model

146 *Governing equation.* In the FEM solver, the nodal displacements are solved as the basic unknown

147 quantity constrained by the equilibrium equation and the compatible condition. Without

148 considering the body force, the macro equilibrium is governed by:

$$149 \begin{cases} \nabla \cdot \boldsymbol{\sigma} = 0 & \text{in } \Omega \\ \boldsymbol{n} \cdot \boldsymbol{\sigma} = \boldsymbol{t} & \text{on } \partial\Omega \end{cases} \quad (1)$$

150 where $\nabla \cdot \boldsymbol{\sigma}$ is the gradient of the stress tensor, \boldsymbol{n} is the unit vector of the outer normal of the

151 boundary, \boldsymbol{t} is the boundary traction, and Ω and $\partial\Omega$ are the domain and the boundary of the

152 macro model, respectively. The governing equation can be written in the weak form as:

$$153 \int_{\Omega} \boldsymbol{\omega} \nabla \cdot \boldsymbol{\sigma} d\Omega = 0 \quad (2)$$

154 where $\boldsymbol{\omega}$ is the test function that is often chosen as the shape function in the Galerkin method.

155

156 *Discretisation.* The domain is discretised into elements where the shape functions N are defined.

157 With integration by parts and some routine operations, Eq. (2) is changed to:

$$158 \int_{\partial\Omega} N^T \boldsymbol{t} d\Gamma - \int_{\Omega} \boldsymbol{B}^T \boldsymbol{\sigma} d\Omega = 0 \quad (3)$$

159 where $\boldsymbol{B} = (\nabla N + \nabla N^T)$. Or simply as:

$$160 \boldsymbol{T} - \boldsymbol{F} = 0 \quad (4)$$

161

162 with the external traction vector $\mathbf{T} = \int_{\partial\Omega} \mathbf{N}^T t d\Gamma$ and the internal force vector $\mathbf{F} = \int_{\Omega} \mathbf{B}^T \boldsymbol{\sigma} d\Omega$.

163

164 *Nonlinear solution.* Due to the plasticity of granular materials, nonlinear iteration should be
165 introduced to obtain the solution. For a given initial solution \mathbf{u}_0 , the internal force \mathbf{F} as a
166 function of the solution \mathbf{u} can be approximated as

$$167 \quad \mathbf{F}(\mathbf{u}) = \mathbf{F}(\mathbf{u}_0) + \mathbf{K}\Delta\mathbf{u} \quad (5)$$

168 where $\mathbf{F}(\mathbf{u}_0) = \int_{\Omega} \mathbf{B}^T \boldsymbol{\sigma}_0 d\Omega$, and \mathbf{K} is the stiffness matrix with a general expression:

$$169 \quad \mathbf{K} = \int_{\Omega} \mathbf{B} \mathbf{D} \mathbf{B}^T d\Omega \quad (6)$$

170 where the matrix \mathbf{D} depends on the material constitutive model used. Then the displacement
171 increment $\Delta\mathbf{u}$ can be obtained by solving

$$172 \quad \mathbf{K}\Delta\mathbf{u} = \mathbf{R}(\mathbf{u}) \quad (7)$$

173 where $\mathbf{R}(\mathbf{u}) = \mathbf{T} - \mathbf{F}$ is the residual force vector of the macro model.

174

175 The matrix \mathbf{D} can be a tangent or secant operator, depending on which solution procedure is used,
176 but the former may lead to a faster convergent rate. In the Newton-Raphson method, the tangent
177 operator \mathbf{D} is needed to construct the tangential element stiffness matrix in order to achieve a 2nd
178 order convergence. In multi-scale calculations, the perturbation method [21] is often used to
179 calculate \mathbf{D} . In Guo's FEM-DEM framework [53], an approximate secant elastic matrix is used.

180

181 *2.2 Micro model*

182 In the FEM-DEM framework, a particle assembly solved by DEM serves as the RVE at each
183 Gauss point to capture the local material responses. The RVE used in the current work has 500
184 particles with their diameters uniformly distributed from 0.006 to 0.014. This particle number is
185 chosen without conducting a convergence test in terms of accuracy for the RVE but is in line with
186 some previous studies. For instance in [22], 400 particles are used in the RVE which seems
187 sufficient to reproduce the homogenisation of the contact fabric tensor at the initial state. More
188 importantly, because the accuracy of the FEM-DEM framework is not the main concern, but rather
189 the machine-learning representation effectiveness of the FEM-DEM model, so choosing a

190 reasonably large number of particles for each RVE should suffice. The boundary condition
 191 (deformation) for each RVE is imposed based on the macro strain ϵ obtained at the Gauss point
 192 by the macro FEM solver. Then the corresponding stress σ and matrix \mathbf{D} are derived from the
 193 RVE via DEM and passed to the FEM solver to update the macro solution.

194

195 The periodic boundary and Hertz's contact model are employed in the DEM simulation. The
 196 normal and tangential contact forces between two contacting particles are calculated as:

$$197 \quad \begin{cases} \mathbf{f}_n = k_n \mathbf{u}_n^c \\ \mathbf{f}_t = k_t \mathbf{u}_t^c \end{cases} \quad (8)$$

198 where \mathbf{u}_n^c and \mathbf{u}_t^c are the relative normal and tangential displacement vectors between the two
 199 particles; and k_n and k_t are the normal and tangential stiffness respectively and defined by

$$200 \quad \begin{cases} k_n = \frac{G}{1-\mu} \sqrt{2r^* u_n^c} \\ k_t = \frac{2G}{2-\mu} \sqrt{2r^* u_n^c} \end{cases} \quad (9)$$

201 where G and μ are the shear modulus and Poisson's ratio of the particle material respectively;
 202 the relative normal displacement is commonly referred to as the overlapping of the two particles;
 203 $r^* = (2r_1 r_2) / (r_1 + r_2)$ is the equivalent radius of the particles with r_1 and r_2 being their radii.

204

205 In the multi-scale simulation, the contact forces in an RVE should be transformed into the average
 206 stress tensor. The homogenised stress tensor of the particle assembly is calculated as [54–56]:

$$207 \quad \sigma = \frac{1}{2V} \sum_{c=1}^{N_c} (\mathbf{f}^c \otimes \mathbf{d}^c + \mathbf{d}^c \otimes \mathbf{f}^c) \quad (10)$$

208 where V is the total volume of the particle assembly, N_c is the total number of contacts within
 209 the volume, \mathbf{f} and \mathbf{d} are the contact force vector and the branch vector connecting the centres of
 210 the two contacted particles, respectively.

211

212 In the FEM-DEM framework, the matrix \mathbf{D} is statistically approximated from the contacts and
 213 homogenised elastic modulus of the particle assembly. The analytical form based on the
 214 assumption of a uniform strain field is given by [57]:

215
$$\mathbf{D} = \frac{1}{V} \sum_{c=1}^{N_c} (k_n \mathbf{n}^c \otimes \mathbf{d}^c \otimes \mathbf{n}^c \otimes \mathbf{f}^c + k_t \mathbf{t}^c \otimes \mathbf{d}^c \otimes \mathbf{t}^c \otimes \mathbf{d}^c) \quad (11)$$

216 where \mathbf{n}_c and \mathbf{t}_c are unit normal and tangential directions of contact c , respectively.

217

218 *2.3 FEM-DEM coupling method*

219 The algorithmic procedures of the coupled FEM-DEM solver is given in Algorithm 1. The multi-
 220 scale coupling method can reproduce the micro-scale mechanical property of granular materials at
 221 the macro scale, as macro strain and micro-scale material response obtained from the RVE
 222 simulation are fully exchanged at the Gauss points. However, a large number of DEM simulations
 223 are required which makes this simulation considerably time-consuming [53, 58]. Most of the
 224 computational time is spent on evaluating the micro DEM model for each Gauss point of the
 225 macro model. In addition, there is a large amount of statistical data generated in this simulation,
 226 including sequences of stress, strain, pore ratio, fabric information, and even the details of the
 227 particle packings at every iteration step. It seems that these datasets are underutilised, and most of
 228 them are discarded after the concurrent simulation.

229

230 The machine learning method can be employed to take full advantage of the intermediate data
 231 generated in the concurrent computations and substantially accelerate the multi-scale computation.
 232 The rest of the paper is dedicated to developing a neural network-based constitutive model that
 233 serves as an efficient surrogate for the concurrent DEM simulation.

Algorithm 1 The FEM-DEM solver

Given: Discretised FEM model, and initialised RVE packings $\mathcal{PK}^{(0)}$

$(\boldsymbol{\sigma}^{(0)}, \mathbf{D}^{(0)}) = \mathcal{PK}(\boldsymbol{\epsilon}^{(0)}, \boldsymbol{\varphi}^{(0)})$ ▷ Initialise the stress and matrix \mathbf{D}

For $n = 1, 2, \dots, N$ **do** ▷ Loading step

Apply the boundary condition at step n to the FEM model

Set $m = 0$, $e_u = 1$

Get $\Delta \mathbf{u}^{(n,m)}$ by solving Eq. (7)

While $e_u > e_{tol}$ **do** ▷ Nonlinear iteration

Compute the strain increment $\Delta \boldsymbol{\epsilon} = \mathbf{B} \Delta \mathbf{u}^{(n,m)}$

Apply $\Delta \boldsymbol{\epsilon}$ to the RVE packing $\mathcal{PK}^{(n,m)}$

Evaluate $(\boldsymbol{\sigma}^{(n,m)}, \mathbf{D}^{(n,m)}) = \mathcal{PK}^{(n,m)}(\boldsymbol{\epsilon}^{(n,m)}, \boldsymbol{\varphi}^{(n,m)})$

Update $\mathbf{R}^{(n,m)}$ and $\mathbf{K}^{(n,m)}$ based on $(\hat{\boldsymbol{\sigma}}^{(n,m)}, \hat{\mathbf{D}}^{(n,m)})$ using Eqs. (5) and (6)

Get $\Delta \mathbf{u}$ by solving Eq. (7)

Update $\Delta \mathbf{u}^{(n,m)} = \Delta \mathbf{u}^{(n,m-1)} + \Delta \mathbf{u}$

Compute the error $e_u = \|\Delta \mathbf{u}\| / \|\Delta \mathbf{u}^{(n,m-1)}\|$

$m = m + 1$

end While

Update $\mathbf{u}^{(n)} = \mathbf{u}^{(n-1)} + \Delta \mathbf{u}^{(n,m)}$, $\boldsymbol{\sigma}^{(n)} = \boldsymbol{\sigma}^{(n,m)}$, $\mathbf{D}^{(n)} = \mathbf{D}^{(n,m)}$

end For

235

236 3 The neural network-based constitutive model

237 3.1 Surrogate micro model constructed by neural network

238 Classical constitutive models are mainly developed based on some phenomenological assumptions.

239 It is generally sufficient to use a small set of experimental data to calibrate the parameters in the

240 model. However, due to the improvement of observation and simulation techniques, these

241 constitutive models may become inadequate to incorporate rapidly increasing high-quality data.

242 Unlike conventional constitutive models, machine learning models directly mine constitutive

243 relationships from data, almost without making any assumptions, since neural network models are

244 sufficiently sophisticated to reproduce various mechanical behaviours of the material [29, 59].

245

246 In the FEM-DEM simulation, the particle assembly attached to each Gauss point is saved in the

247 computer RAM for the next loading step to reproduce the history-dependent effect of the granular
 248 material, which is also the inherent reason for its huge memory cost. One of the central challenges
 249 of the machine learning model is to characterise a general loading history with as few variables as
 250 possible. **As the network-based model will no longer gain access to the particle assembly,** proper
 251 internal variables should be included to account for the history-dependent (plastic) state of
 252 granular materials in the FEM-ML framework. **However, it is hard to explicitly calibrate the**
 253 **plastic state, such as plastic work and yield surface, since they cannot be easily derived from the**
 254 **macro DEM simulated results (i.e. strain, stress, void ratio).**

255

256 Unlike traditional internal variables, the history variables used for machine learning can have no
 257 physical meaning and can be merely used to quantify the historical influence or sequence order.
 258 Thus, they only need to possess some very basic mathematical properties, such as monotonicity. In
 259 this work, the accumulation of the absolute strain increments:

$$260 \quad \boldsymbol{\varphi}^{(n)} = \sum_i^n |\Delta \boldsymbol{\epsilon}^{(i)}| \quad (12)$$

261 which is conveniently available and interpretable, is used as the history variable. Subsequently, the
 262 constitutive relationship to be obtained can be expressed as

$$263 \quad \hat{\boldsymbol{\sigma}}^{(n)} = \mathcal{NN}(\boldsymbol{\epsilon}^{(n)}, \boldsymbol{\varphi}^{(n)}) (n = 1, \dots, N) \quad (13)$$

264 where \mathcal{NN} denotes a multi-layer fully connected neural network, the superscript n represents
 265 the sample number and N is the total number of training samples. Then the prediction error is
 266 measured by the MSE (mean square error) as:

$$267 \quad err = \frac{1}{N} \sum_n^N \|\hat{\boldsymbol{\sigma}}^{(n)} - \boldsymbol{\sigma}^{(n)}\|^2 \quad (14)$$

268 **Our experiments show that using the accumulation of the absolute strain increments defined in Eq.**
 269 **(12) as the internal variable seems to be able to encode the unique stress-strain historical state of**
 270 **materials and that the trained network can effectively capture the path-dependent behaviour of**
 271 **granular assemblies.**

272

273 In the FEM-DEM framework, the approximated secant matrix is used as the stiffness matrix
 274 which is derived from the fabric of the particle assembly (i.e. Eq. (11)), parallel with the stress and
 275 other state variables. The requirement of some specific micro-scale information in FEM-DEM

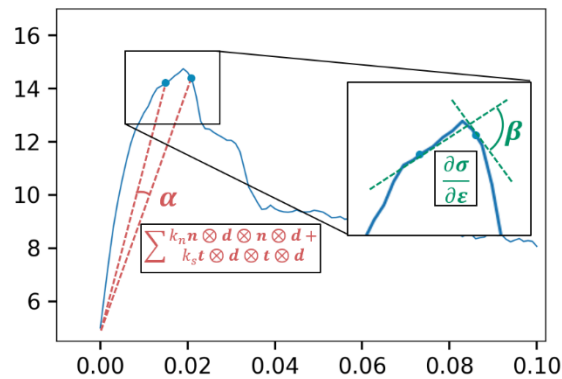
276 calculation, such as the configuration and dynamic properties of the assembly, will demand much
 277 greater computational time and memory resources, dramatically slowing down the multi-scale
 278 scheme. There are various features (such as stress, strain, void ratio, and fabric information)
 279 generated at every time step in the whole FEM-DEM simulation. However, only necessary
 280 features should be involved in machine learning to calculate the operator \mathbf{D} . Simplifying and
 281 encapsulating these features into simple pairs as $(\epsilon, \varphi, \sigma, \mathbf{D})$ will alleviate difficulties in the
 282 network training and accelerate the computation.

283

284 There are mainly two ways to represent the operator \mathbf{D} for the nonlinear iteration in a machine
 285 learning scheme: (1) tangent operator, and (2) secant operator. The tangent operator will be ideal
 286 for the Newton-Raphson solution of the nonlinear problem if the stress-strain function is
 287 sufficiently smooth and differentiable. In machine learning-based work, some computations for
 288 continuous media use the AD (Automatic Differentiation) in Tensorflow [60] to obtain $\mathbf{D} = \frac{\partial \sigma}{\partial \epsilon}$
 289 [28, 35–37] or the automatic differentiation tool AceGen [61] to directly update the stiffness
 290 matrix via $\mathbf{K} = \frac{\partial \mathbf{R}}{\partial \mathbf{u}}$ [62].

291

292 Unfortunately, the stress-strain curves extracted from granular material simulations fluctuate
 293 dramatically, as shown in Figure 2. The network is trained to capture the main trend of these
 294 characteristics but is not intelligent enough to distinguish the valuable constitutive relationship
 295 from the fluctuations. Thus, the tangent operator calculated via the auto-differentiation may
 296 change sharply and largely deviate from the right direction, which will adversely affect the
 297 computational stability.



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Figure 2: Schematic of the approximate secant matrix and the tangent matrix

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The secant matrix can be approximated via Eq. (11) since the contact details in a particle assembly are available in DEM simulation. It is also theoretically available via the machine learning as strain ϵ and history variable ϕ are both input variables. Therefore, in consideration of the computational stability, the approximate secant operator together with the stress is adopted via the \mathcal{NN} to complete the nonlinear iteration:

$$(\hat{\sigma}^{(n)}, \hat{D}^{(n)}) = \mathcal{NN}(\epsilon^{(n)}, \phi^{(n)}) \tag{15}$$

3.2 Procedures of the multi-scale coupling method

The procedure of the FEM-ML framework is outlined in Algorithm 2. There are no particle assemblies that need to be initialised and updated since the neural network will directly calculate the stress and the secant operator after the strain and history variables are fed into the trained network. **Note that the only physical parameter involved in our neural network-based constitutive model is the particle-scale Hertz contact law for discrete element modelling of granular materials. There are no other phenomenological constitutive models involved in the FEM-ML modelling.**

Algorithm 2 The FEM-ML solver**Given:** Discretised FEM model, and well-trained neural network \mathcal{NN}

$$\left(\hat{\boldsymbol{\sigma}}^{(0)}, \hat{\mathbf{D}}^{(0)}\right) = \mathcal{NN}\left(\boldsymbol{\epsilon}^{(0)}, \boldsymbol{\varphi}^{(0)}\right) \quad \triangleright \text{Initialise the stress and matrix } \mathbf{D}$$

For $n=1, 2, \dots, N$ **do** \triangleright Loading stepApply the boundary condition at step n to the FEM modelSet $m=0$, $e_u=1$ Get $\Delta \mathbf{u}^{(n,m)}$ by solving Eq. (7) with $\hat{\boldsymbol{\sigma}}^{(n-1)}, \hat{\mathbf{D}}^{(n-1)}$ **While** $e_u > e_{tol}$ **do** \triangleright Nonlinear iteration

Compute $\boldsymbol{\epsilon}^{(n,m)} = \boldsymbol{\epsilon}^{(n-1)} + \mathbf{B}\Delta \mathbf{u}^{(n,m)}$

Evaluate $\left(\hat{\boldsymbol{\sigma}}^{(n,m)}, \hat{\mathbf{D}}^{(n,m)}\right) = \mathcal{NN}\left(\boldsymbol{\epsilon}^{(n,m)}, \boldsymbol{\varphi}^{(n-1)}\right)$

Obtain $R^{(n,m)}$ and $K^{(n,m)}$ from $\left(\hat{\boldsymbol{\sigma}}^{(n,m)}, \hat{\mathbf{D}}^{(n,m)}\right)$ via Eqs. (5) and (6)Get $\Delta \mathbf{u}^{(n,m)}$ by solving Eq. (7)

Calculate $e_u = \|\Delta \mathbf{u}^{(n,m)} - \Delta \mathbf{u}^{(n,m-1)}\| / \|\Delta \mathbf{u}^{(n,m-1)}\|$

 $m = m + 1$ **end While**

Update $\mathbf{u}^{(n)} = \mathbf{u}^{(n-1)} + \Delta \mathbf{u}^{(n,m)}$, $\boldsymbol{\sigma}^{(n)} = \boldsymbol{\sigma}^{(n,m)}$, $\mathbf{D}^{(n)} = \mathbf{D}^{(n,m)}$

end for

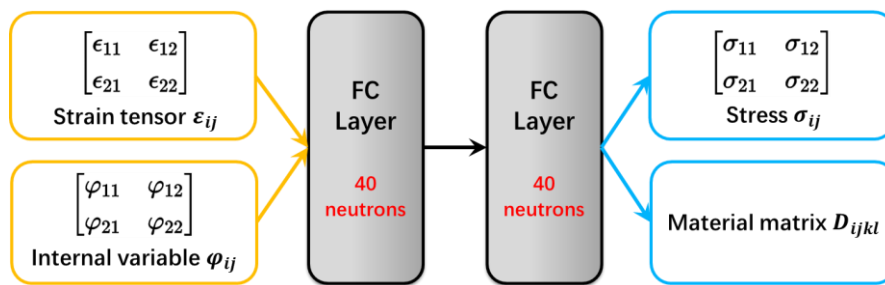
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318 *3.3 Network architecture*

319 Most of the work in machine learning, especially deep learning, uses fairly complex network
 320 structures [37, 38, 49, 63], while traditional phenomenological constitutive models of granular
 321 materials generally use around ten parameters to approximately describe the stress-strain
 322 relationships. As shown in [64], PINNs (Physics-constrained neural networks) constructed by
 323 parsimonious networks can exhibit fairly good prediction accuracy. Using simpler networks can
 324 also largely reduce the number of parameters involved. Additionally, we found that simple neural
 325 networks are more conducive to computational stability, which may be related to the fact that the
 326 predicted values given by a simple network tend to be smoother. By considering the simplicity of
 327 deployment, computational stability and efficiency, a simple multilayer fully connected network is
 328 chosen in our work to be integrated with FEM calculation, with the expense of slightly lower
 329 training accuracy.

330

331 Figure 3 shows the neural network used, which is composed of two fully connected hidden layers,
 332 each having 40 neural nodes (or neurons). The ReLU (Rectified Linear Unit) activation function is
 333 adopted except for the output layer (without activation function). The input layer includes the
 334 current strain and history variables, and the output layer produces the corresponding stress and D
 335 matrix. The Adam algorithm (with $lr = 1e-3$, $betas = (0.9, 0.999)$, $eps = 1e-8$, $weight_decay = 0$) is
 336 employed to optimise the weights and biases in the training process. PyTorch is used to implement
 337 the neural network. Note that no attempt has been made to optimise the network in terms of the
 338 number of hidden layers and neurons in each layer.



339

340 Figure 3: Architecture, input and output features of the neural network

341 3.4 Sampling

342 In general, machine learning models are good interpolators but often do not perform well for
 343 extrapolation. To develop a machine-learning-based constitutive model suitable for a wide range
 344 of strain paths, one solution is to accommodate as many strain-stress pairs as possible, so that all
 345 the “extrapolation” become “interpolation”. This, however, imposes two challenges. Firstly, it will
 346 be extremely challenging to generate sufficient training datasets to fully cover all the possible
 347 strain-stress paths for a certain type of granular material. The task becomes practically impossible
 348 when further considering all possible loading-unloading combinations. This is a sharp contrast to
 349 the conventional constitutive model where the model parameters can be calibrated by a relatively
 350 small number of conventional triaxial tests in geomechanics. Secondly, the presence of a large
 351 number of datasets is also a challenge for the training of neural networks—a large number of
 352 datasets demand expensive computational costs. Thus, it is important to utilise any useful
 353 condition to increase the effectiveness of the training dataset, leading to the reduction of sampling
 354 and training costs.

355

356 For homogeneous materials, some symmetric properties can be exploited for reducing material
357 sampling requirements in the data-driven paradigm [39, 44]. For instance, both strain and stress
358 tensors at any data point pair can be mapped to their principal components via a proper rotation.
359 With this principal space mapping, the dimensions of the strain/stress sampling space are reduced
360 from six to three [39, 63]. In Tang’s work [65–67] the 3D sampling space is further reduced to a
361 1D space. Note, however, that for granular materials, the strain and stress matrix may not be
362 simply mapped to the principal space because the principal directions of the strain and stress start
363 to deviate since plasticity (or non-affine deformation) emerges [8, 68]. This non-coaxial issue is
364 considered in our work by casting all components of the strain and stress tensor into the network
365 model.

366 *3.4.1 Sample generation*

367 Gaussian process is introduced to generate smooth random loading paths in order to cover a large
368 sampling space of (ϵ, σ) in our work. In some machine learning frameworks [38, 49], Gaussian
369 process [69] is used in the micro-scale random loading path preparation. However, the DEM
370 simulations may return unreasonable results when the granular materials are over-compressed or
371 pulled up. So, this work applies the randomly generated loading paths as macro boundary
372 conditions instead of the micro-scale RVE. The computed strain at each Gauss point of the macro
373 model is applied as the boundary condition to the corresponding micro-scale RVE. In this way, a
374 large number of samples closely related to the solution space can be directly generated. Applying
375 the boundary condition directly to the macro model is also much simpler, as there is no need to
376 check the compatibility of the generated results.

377

378 The macro model used to generate training data is a simple biaxial test discretised by three
379 different finite element meshes. The detail will be described in Section 4.1 and shown in Figure 6.
380 A series of FEM-DEM biaxial simulations subject to randomly constructed boundary loading
381 paths is carried out to generate training data. To avoid over-compression or stretching, the random
382 path is only applied to control the transverse confining pressures for the biaxial compression test.
383 All the REV particle models are initially compressed by an isotropic confinement pressure of
384 100kPa. Further details will be given in Section 4.

385

386 The Gaussian process defines a distribution function $f(\mathbf{x})$ that is completely specified by its
387 mean function $m(\mathbf{x})$ and covariance kernel function $\kappa(\mathbf{x}, \mathbf{x}')$, i.e., a Gaussian process can be
388 represented as:

$$389 \quad f(\mathbf{x}) \sim \mathcal{GP}(m(\mathbf{x}), \kappa(\mathbf{x}, \mathbf{x}')) \quad (16)$$

390 where the mean value $m(\mathbf{x})$ is kept to be the initial confinement pressure of 100 kPa. The
391 smoothness of the generated sequence can be controlled by specifying the kernel function.
392 The squared distance function is used as the covariance kernel function and described by:

$$393 \quad \kappa(\mathbf{x}, \mathbf{x}') = \exp(-v_c \theta_k(\mathbf{x}) \|\mathbf{x} - \mathbf{x}'\|^2) \quad (17)$$

394 where variable $x \in [0,1]$ can be regarded as the pseudo-time to control the loading step, and its
395 discretised values $\mathbf{x} = [0.01, 0.02, \dots, 1]$, corresponding to 100 loading steps, are used as the vector
396 for kernel κ generation; $v_c \in [1,5]$ is a coefficient used to control the curvature of the random
397 path (a larger v_c narrows the band of the kernel, but increases the curvature of the stress curve, as
398 illustrated in Figure 4); and $\theta_k(\mathbf{x})$ is a linear function of \mathbf{x} starting from 0 to ensure that the
399 random confining pressure starts from the initial consolidation pressure of 100 kPa.

400 The Gaussian process outlined above is used to generate ten random loading paths for the
401 confinement pressure for each finite element model (Figure 6). The total number of samples is
402 summarised in Table 1, with an overall sum of nearly 17 million data points generated. Due to the
403 nonlinear macroscopic mechanical properties of granular materials, each loading step needs
404 nonlinear iteration. On average, a loading step takes about 20 iterations. Therefore, more than
405 2000 calculation results are generated in 100 loading steps.

406

407

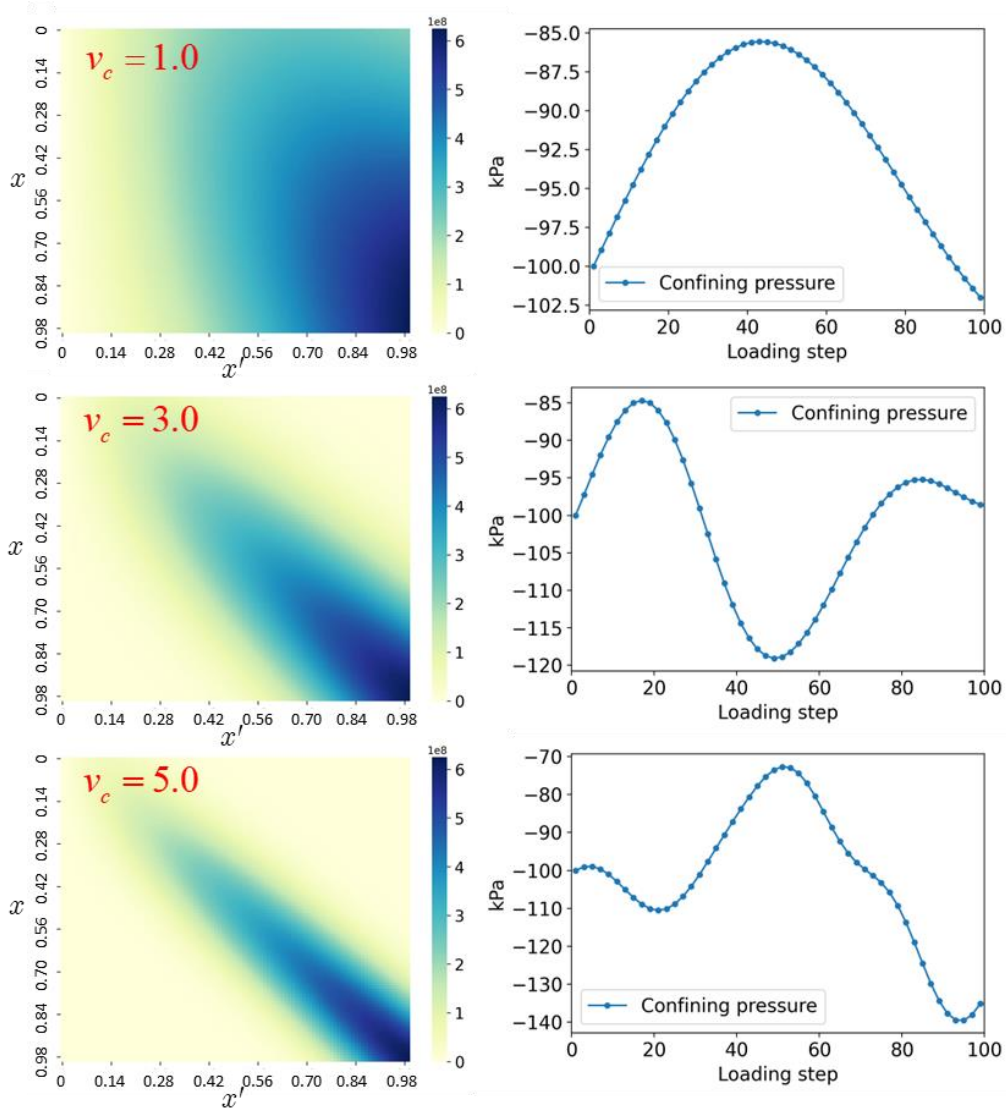
408

Table 1 Summary of the training samples generated at three mesh levels

Loading path number	Coarse	Medium	Fine
0	76064	308608	1295360
1	75456	316544	1243136
2	75264	308736	1314304
3	81696	309248	1303040
4	77984	304768	1312768
5	77472	317440	1304576
6	76128	303744	1288704
7	75296	310016	1285632
8	75040	313472	1296896
9	75456	331904	1330176
Total training samples		16864928	

409

410



411

412 Figure 4: Random loading path generation via Gaussian process: the kernel functions

413

(left) and random paths (right) for different ν_c values

414

415 *3.4.2 Active learning-based resampling.*

416 It may not be a good idea to use all the available data points, particularly in huge datasets, for

417 network training. Using all data points may prevent the optimizer from finding the most

418 informative datasets for the targeted training as there may be many data points that are repetitive

419 or very similar. In this work, we adopt an active learning strategy to effectively select samples

420 from our prepared large datasets.

421

422 The active learning method [70] is introduced for effective resampling from a dataset with
 423 massive data points. The key idea behind is that a machine learning algorithm can achieve greater
 424 accuracy with fewer training samples if the algorithm is facilitated to choose essential data points
 425 according to what it has learned. In addition to resampling, active learning can also guide sample
 426 generation, especially when generating a sample is expensive. The active learning method is able
 427 to evaluate the network performance at a data point (i.e. strain and internal variable) without
 428 knowing the output/response (i.e. stress) and thus can determine whether the data point is valuable
 429 to be generated for network training.

430

431 The active learning procedure used is as follows. Firstly, a selected number of networks with the
 432 same architecture and hyperparameters but different randomly generated initial weights and biases
 433 are trained on partial datasets or smaller datasets generated based on a subset (i.e. in the coarse FE
 434 mesh here). Subsequently, these trained networks are used to make predictions at different data
 435 points (i.e. in the fine FEM mesh), and then evaluate their uncertainty levels ξ , defined to be the
 436 standard deviation of all the predictions at each datapoint. The process can be summarised as:

$$437 \left\{ \begin{array}{l} \hat{\mathbf{y}}^{(n)} = \frac{1}{m} \sum_{i=1}^m \mathcal{NN}_i(\boldsymbol{\sigma}^{(n)}, \boldsymbol{\varphi}^{(n)}) \\ \xi^{(n)} = \sqrt{\frac{\sum_{i=1}^m \|\mathcal{NN}_i(\boldsymbol{\sigma}^{(n)}, \boldsymbol{\varphi}^{(n)}) - \hat{\mathbf{y}}^{(n)}\|^2}{m-1}} \end{array} \right. \quad (18)$$

438 where m is the number of the randomly trained networks \mathcal{NN}_i ($i=1,2,\dots,m$), $\hat{\mathbf{y}}^{(n)}$ is the
 439 averaged output of all the networks at a selected Gauss point n in the fine-meshed model. After
 440 evaluation, points with an uncertainty value higher than a certain level will be added to the
 441 training samples to retrain the models until a satisfactory level of uncertainty is achieved for all
 442 the Gauss points concerned. Then one of the trained models is chosen to be used for prediction in
 443 the FEM-ML solver. Thus the active learning in the present work acts as a detector to seek the
 444 locations where the randomly trained model performs poorly, i.e. the data points are
 445 underrepresented yet. This active learning resampling is referred to as the uncertainty-level based
 446 active learning scheme.

447 In addition to the above uncertainty-level base active learning scheme, some other indicators
 448 can also be utilised to guide the resampling process. An alternative, equivalent shear strain-based

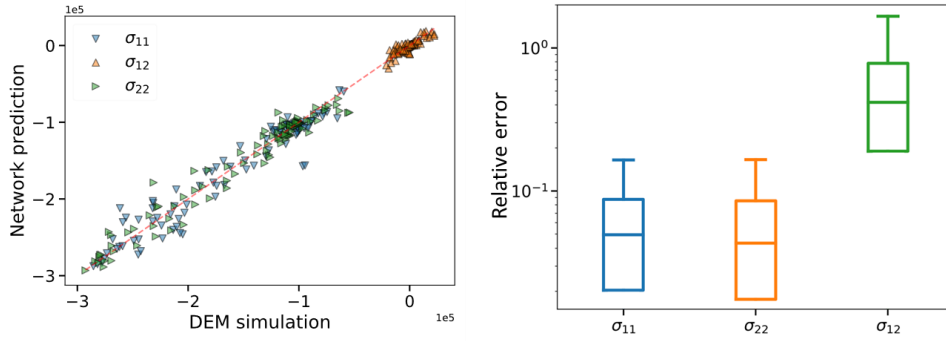
449 resampling scheme is another possible option for the problems concerned in Section 4. The details
450 of using this option and comparison with the uncertainty-level base scheme are given in Section
451 4.1.2.

452 3.5 Training

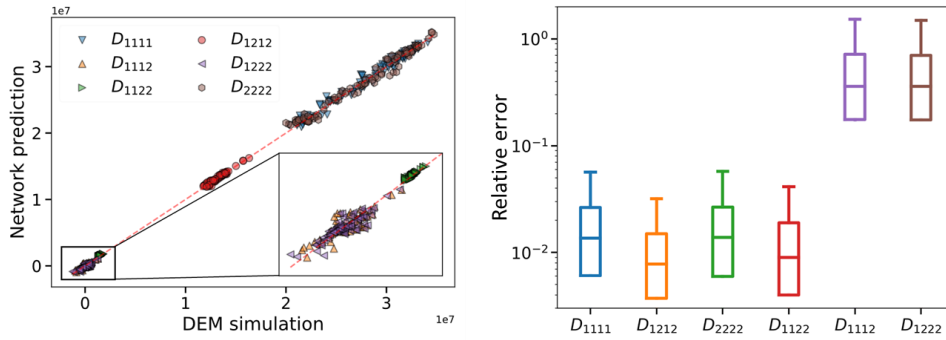
453 The neural network is trained on the dataset collected from a random Gaussian process controlled
454 biaxial loading. The datasets are split into training, validating, and testing sets weighing 70%, 15%,
455 and 15%, respectively. The early stopping technique [71] is utilised in the training process to avoid
456 overfitting and thus maximise the generalisation ability of the ML model. The MSE (mean square
457 error) of both training and validation datasets is evaluated every ten epochs. The training process
458 is ceased once the MSE of the validation dataset has no improvement after 1,000 training epochs.

459

460 After training, the prediction ability of the model and the prediction error are checked, as shown in
461 Figure 5. The trained \mathcal{NN} model, in most cases, is competent to predict with satisfactory
462 accuracy, especially in the prediction of the two stress components (σ_{11} and σ_{22}) and four
463 components ($D_{1111}, D_{1122}, D_{1212}, D_{2222}$) of the material matrix. However, the \mathcal{NN} model seems
464 not to perform well in predictions of σ_{12} , D_{1112} and D_{1222} , as shown in the right column of
465 Figure 5 where the relative errors reach around more than 100 per cent. Nevertheless, this is
466 because, in an isotropic hyper-elastic model or quasi-elastic stage of the granular material, the
467 components of D_{1112} and D_{1222} should be zero [72] and σ_{12} is also close to zero. Thus, it is more
468 appropriate to check the absolute error for these components. The left column of Figure 5 shows
469 that their predicted values are indeed close to zero, indicating their prediction accuracy is also
470 satisfactory.



(a) Prediction of the stress vector



(b) Prediction of the material matrix

471

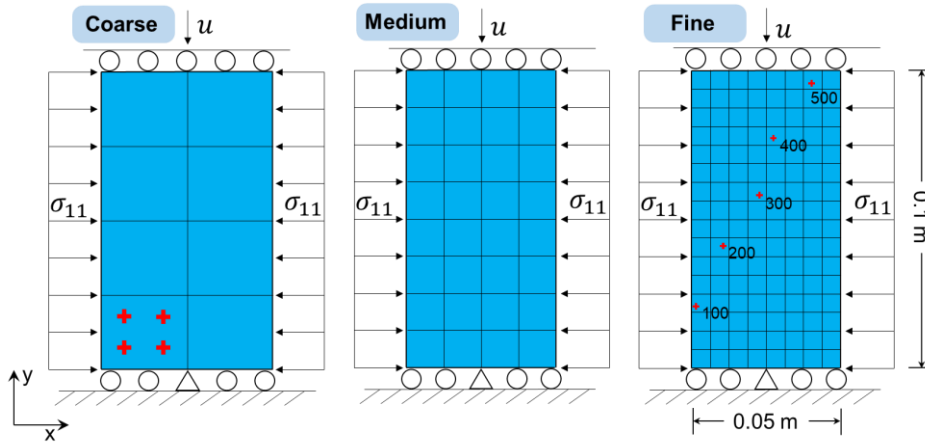
472 Figure 5: ML prediction versus the results of the DEM simulation – left column:

473

comparison; right column: relative error.

474 4 Computational examples

475 4.1 Baseline problem - the biaxial compression test



476

477 Figure 6: Configuration of the biaxial compression test and three meshes

478 The biaxial compression test, shown in Figure 6, is simulated as the baseline to validate the

479 proposed FEM-ML framework. Three different meshes are used to discretise the problem domain:

480 coarse (2×4 elements), medium (4×8 elements) and fine (8×16 elements) meshes, with each

481 element having 4 Gauss points.

482 Table 2 Parameters of the granular materials

Density (kg/m ³)	Young's modulus (MPa)	Poisson's ratio	Friction angle (rad)	Damping ratio
2650	600	0.8	0.5	0.1

483

484 Initially, the vertical displacement-controlled loading is linearly applied to the top boundary of the
485 biaxial model until the macro vertical strain reaches 0.1. The Gaussian process outlined in Section
486 3 is used to generate ten random loading paths to control the confinement pressure for each
487 meshed model. Parameters used in the lower-scale DEM simulation are shown in Table 2. Both
488 the FEM-DEM and the FEM-ML simulations are carried out for the coarse and fine meshes with
489 an identical loading/boundary condition to assess whether our proposed FEM-ML framework is
490 capable of reproducing the multiscale mechanical response and effectively accelerating the
491 computation. The medium mesh is used later in Section 4.1.4.

492

493 *4.1.1: Comparison of the prediction results with different meshes.*

494 To comprehensively compare the prediction capability of the proposed FEM-ML framework, six
495 cases that combine the two meshes (coarse and fine) for both the training dataset generations and
496 the FEM-ML simulations are considered and listed in Table 3. The simulated results are displayed
497 in Figure 7 (top force) and Figure 8 (displacement field).

498

499 Table 3 The cases using different meshes for training and FEM-ML simulations

Case	Mesh used for obtaining training samples	Mesh used for FEM-ML simulations
A	Coarse	Coarse
B	Fine	Coarse
C	Hybrid	Coarse
D	Fine	Fine
E	Coarse	Fine
F	Hybrid	Fine

500

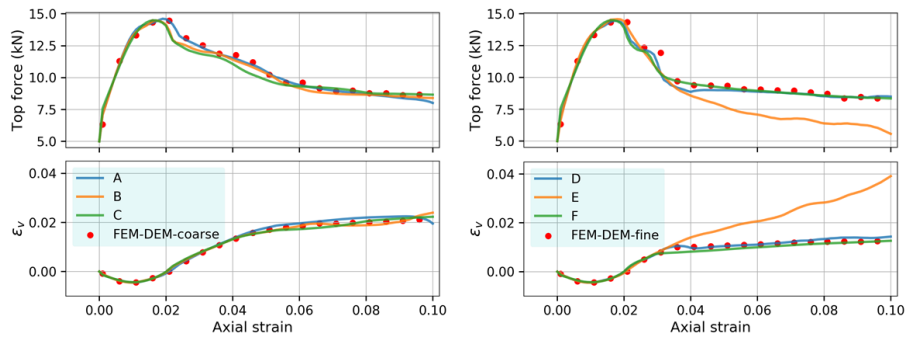
501 These two figures illustrate that all results agree well with the DEM simulations except for
502 Case E. The shear band simulated with the fine mesh is narrower than that with the coarse
503 mesh in the FEM-DEM simulations. The final deformed configuration of Case E is more

504 similar to Case A instead of the fine-meshed FEM-DEM simulation. In Cases A and E, only
 505 datasets collected from the coarse mesh simulations are employed to train the neural network. The
 506 Gauss points are clearly sparser in the coarse mesh, thus there are not sufficient points to represent
 507 the strain-stress relationship in the fine mesh.

508

509 On the other hand, due to the neural network's good interpolation ability, all of the stresses and
 510 tangent matrices in the shear bands in the fine mesh can be well approximated by the network
 511 trained on the coarse datasets. Although the fine mesh is used for the macro-level simulation in
 512 Case E, the strain-stress relationship is still derived from the coarse mesh datasets. Therefore, the
 513 shear band obtained in Case E still looks like a duplicate of Case A.

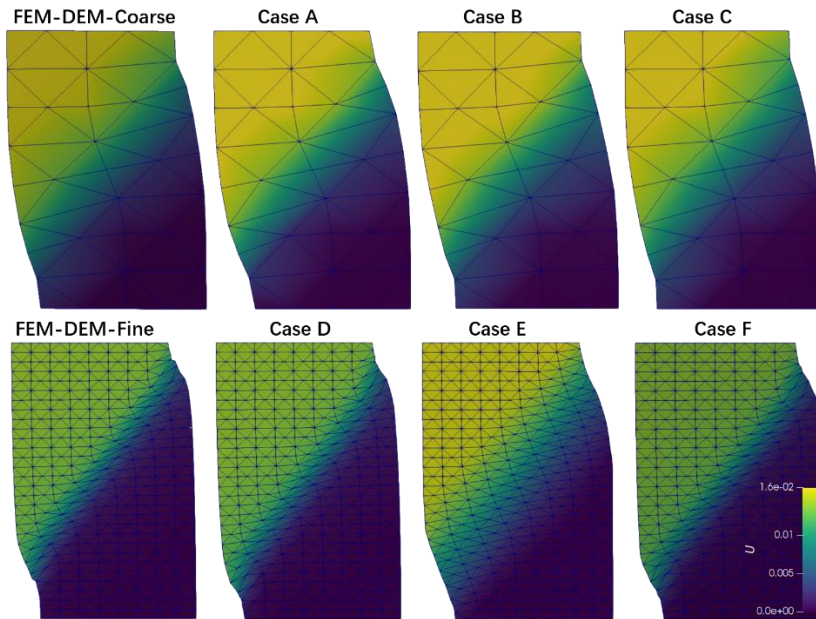
514



515

516

Figure 7: The comparison of the top force between different cases



517

518

Figure 8: The final deformed configurations of the different cases

519

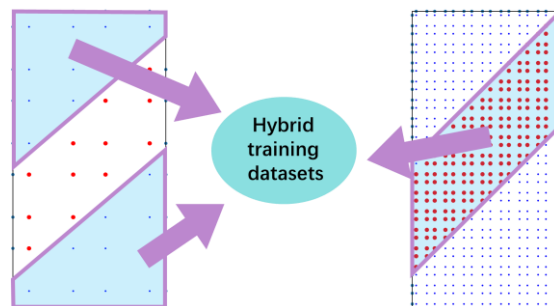
520 Due to the fact that the strain distributions in the upper and lower regions of the domain are very
521 similar for both coarse and fine meshes, attentions are focused on the shear band obtained from
522 different cases. Cases C and F are subsequently carried out to investigate this issue, where the
523 network models are trained on datasets gathered through a hybrid sampling method as shown in
524 Figure 9. In this hybrid sampling method, the data points in the shear band are obtained from the
525 fine mesh simulations, while the data points in the upper and lower triangular regions are collected
526 from the coarse-meshed simulations.

527

528 Figure 8 shows that the final deformed configurations of Cases C and F agree well with the FEM-
529 DEM simulations. This suggests that the training datasets in the shearing band significantly
530 influence the performance of the purely data-based model. Only with the proper training datasets,
531 can the network model work well and attain the expected results.

532

533



534

535 Figure 9: Collecting the training datasets in a hybrid way

536

537 4.1.2 Automatic resampling via active learning.

538 The two hybrid cases (Cases C and F) indicate that the significance of different data points varies
539 greatly. The points in the shear band play a significant role in the current problem. In contrast,
540 constitutive relationships contained by points in the upper and lower triangles of the fine-meshed
541 model can be easily represented by the data points from the coarse-meshed model. To fully utilise
542 these features, the uncertainty-level-based active learning scheme introduced in Section 3.4.2 is
543 used for automatic resampling. The detailed procedure is outlined in Figure 10 and explained

544 below:

- 545 1) Choose five separate networks with the same architecture and hyperparameters as described in
- 546 Section 3.3 but with different randomly initialised weights and biases which are pre-trained
- 547 based on the datasets obtained from the coarse mesh.
- 548 2) Use the five pre-trained models to predict the stresses and \mathbf{D} matrices for all strains of the
- 549 data points collected from the fine mesh FEM-DEM simulations.
- 550 3) Evaluate the prediction uncertainty level at each data point based on the five predicted stresses
- 551 by Eq. (18).
- 552 4) Add 30% of the data points that have the highest uncertainty levels to the training samples
- 553 used in the pre-training phase.
- 554 5) Re-train one pre-trained model on the enriched dataset and use it as the network model for the
- 555 final FEM-ML simulation.

556

557 Note that the number of NNs used in the above active learning resampling cannot be determined

558 analytically but is a result of trial and error in practice. The main principle is that we wish to use

559 as few NN models as possible to find a reliable variance ranking which can effectively recognise

560 the most informative points to improve the current NN predictions. To strike a balance between

561 accuracy and computational costs, 5 is found to be a satisfactory number of the required networks

562 in the current work.

563

564 Figure 10 clearly illustrates that the prediction uncertainty is prominent in the shear band,

565 indicating that the data points in the shear band have a stronger influence on the network

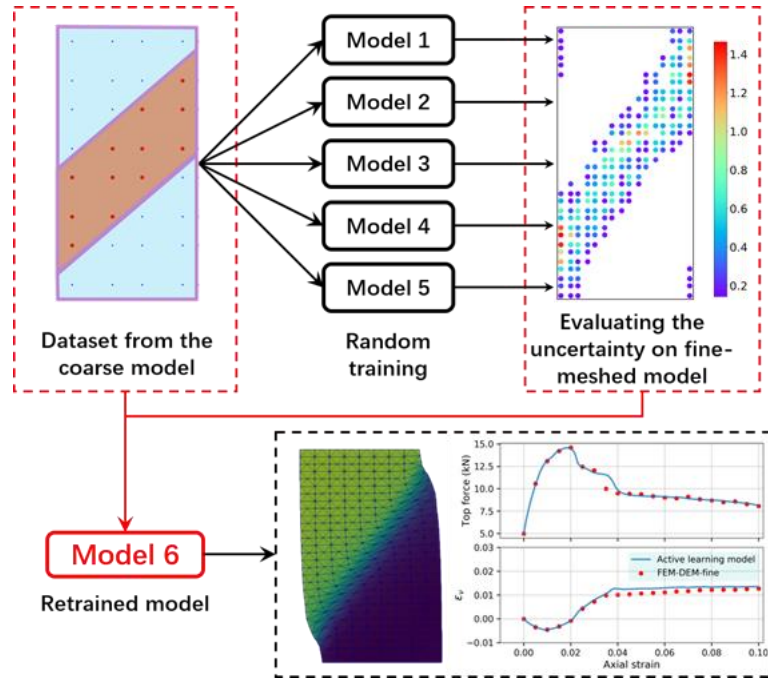
566 prediction. Locations of the points with high uncertainty levels agree well with the red data points

567 extracted on the right side of Figure 9, which explains why the result of Case F is improved. Thus,

568 active learning can be used as a filter to identify the locations where the trained network model

569 performs poorly and samples at these locations should be generated or added to enrich the training

570 dataset.



571

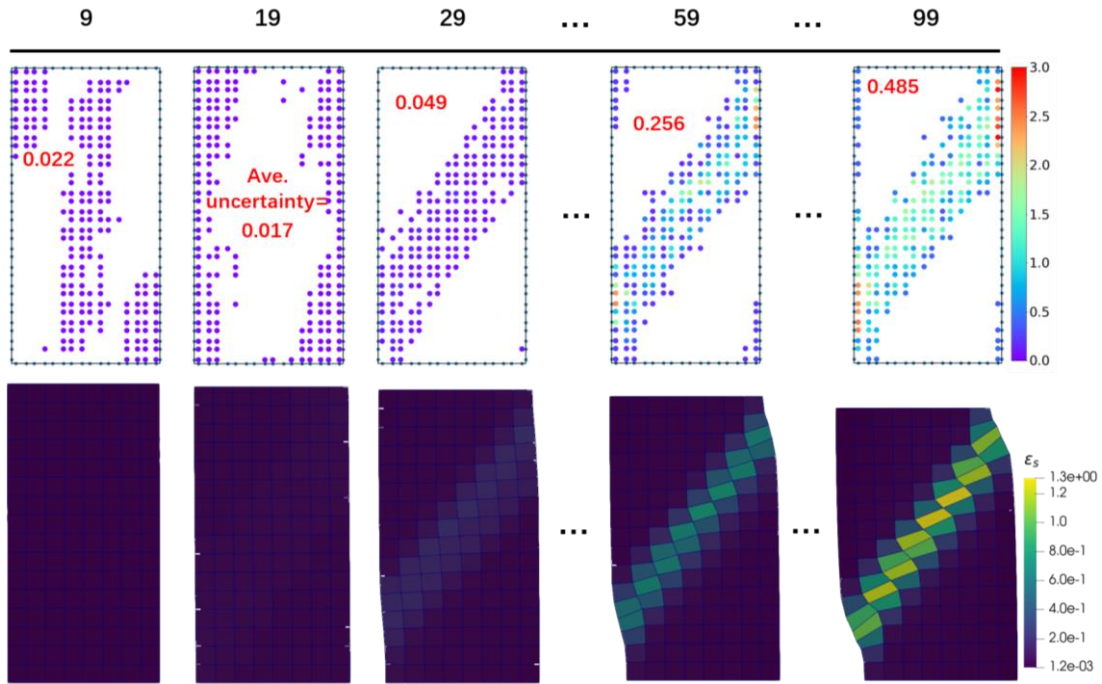
572 Figure 10: The flowchart of the model training process based on the active learning
 573 resampling

574

575 As the newly added data points in the above active learning resampling process are almost all
 576 located in the shear bands of the deformed configuration, it indicates that the magnitude of shear
 577 strain may be utilised as an alternative indicator for the resampling purpose. To investigate the
 578 possibility of using the equivalent shear strain as the indicator for active learning resampling, we
 579 compute the equivalent shear strains at all Gauss points at each load step. Then 30% of the data
 580 points having the highest uncertainty levels at several load steps are depicted on the top row in
 581 Figure 11 (where the average uncertainty values are also given), while the bottom row shows the
 582 corresponding contour plots of the equivalent shear strain.

583 The figure shows that the shear band (or strain localisation) starts to emerge at around load
 584 step 29 and fully develops from step 59 onwards. The corresponding distributions of the Gauss
 585 points with 30% of the highest shear strain for these load steps are indeed around the shear band
 586 and these coincide with the prediction by the uncertainty-level based active learning scheme
 587 shown in Figure 10. Thus it seems reasonable to conclude that the (equivalent) shear strain can be
 588 used as an alternative indicator for the active learning resampling. However, at the beginning of
 589 the loading and well before the shear band is formed, the strain field is nearly uniform across the

590 whole domain. Consequently, the shear strain indicator is unable to distinguish the difference
 591 between data points over the whole domain, and the selected data points may not be ideally
 592 located. It is clearly displayed for the first two load steps where the selected points are not
 593 uniformly distributed across the domain.



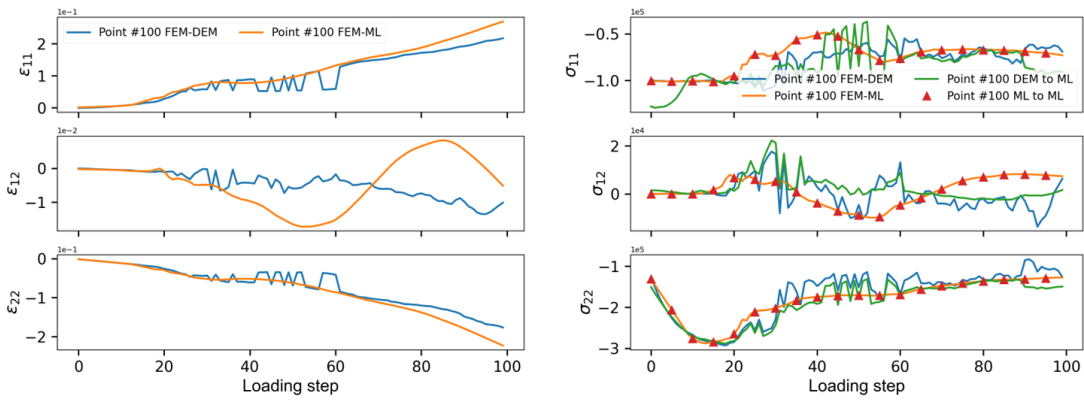
594
 595 Figure 11 The shear strain based active learning resampling at seven load steps: Top
 596 row – 30% of the data points with highest uncertainty levels (the number in red is
 597 their average value); Bottom row – equivalent shear strain distribution

598
 599 In conclusion, the shear strain can serve as a simple alternative indicator for resampling
 600 where large shear strain or shear localisation may be a dominant feature, but it may not be an
 601 effective indicator when the shear strain difference is small over the problem domain. More
 602 importantly, the shear strain indicator is problem-specific and hence its applicability is rather
 603 limited. For different problems, we may need to seek different problem-specific indicators for
 604 resampling in an indicator-based sampling scheme. On the contrary, the uncertainty-level based
 605 active learning may be slightly more complex in terms of usage, but it is generic and completely
 606 based on the evaluation of the uncertainty level of prediction results without any prior knowledge
 607 of the datasets concerned.

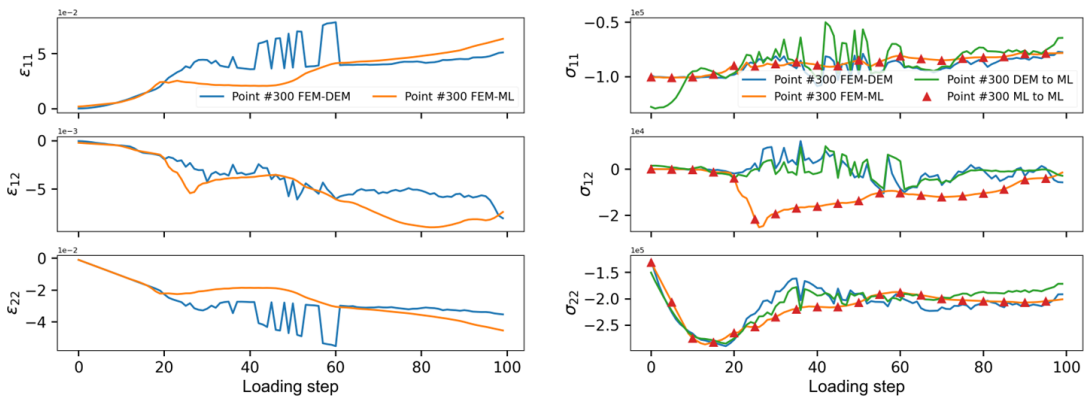
608

609 4.1.3 Comparison of strain-stress predictions at Gauss points

610 To further assess the validity of the new proposed framework, the strain-stress responses obtained
 611 by various models at Gauss point level are checked. The two Gauss points, #100 and #300, whose
 612 locations are shown in Figure 6, are chosen for inspection. In the right column of Figure 12, the
 613 blue line is the stress evolution history from the FEM-DEM simulation. The green line is the
 614 predicted stress from the trained network model fed with the strain obtained from the FEM-DEM
 615 simulation. The orange line is the stress from the FEM-ML simulation. The triangles denote the
 616 stress predicted by the network model using the strain obtained from the FEM-ML simulation.



(a) Gaussian point #100



(b) Gaussian point #300

617
 618 Figure 12: Comparison of the local strain and stress responses from different solution
 619 schemes

620 The FEM-ML framework works well in predicting the main trends of the macroscopic stress
 621 response of granular materials. However, the trained network model inherently produces a smooth
 622 output but cannot reproduce a more oscillating output induced by the transient nature of DEM
 623 simulations, as notably visible in the FEM-DEM results. In FEM-DEM simulations, the strain
 624 updated via $\Delta \epsilon = \mathbf{B} \Delta \mathbf{u}$ will also oscillate because the DEM-induced stress fluctuation is

625 passed on to the FEM solver, as shown in the left column of Figure 12.

626

627 It is worth noting that the fluctuating or noisy output is the main obstacle in network model
628 training, especially in high-dimensional problems, since it will aggravate the difficulty to
629 distinguish useful information from the noise. Large and sharp fluctuations make the predicted
630 curve nondifferentiable, which is also the reason that the automatic differentiation method
631 available in most machine learning libraries cannot be directly used to obtain tangent operators in
632 granular material simulation.

633

634 The network prediction directly using the strain obtained from the FEM-DEM simulation (green
635 line) seems to have fully captured the fluctuation of the RVE results, except for the initial stress
636 predictions of σ_x . After further inspection, we find that the initial strain returned by the FEM-ML
637 solver is slightly different from the FEM-DEM solver due to the error in the predicted stress at the
638 initial state. As the stress component σ_x is supposed to be near 100 kPa over the whole course of
639 loading, its error in the figure is relatively more visually obvious than the other two.

640

641 4.1.4 Performance comparison

642 It is important to evaluate the efficiency of the FEM-ML framework. The performance of both
643 FEM-DEM and FEM-ML simulations for Case D is compared without including the active
644 learning resampling on a laptop computer (i5-8500 6 Cores@3.00GHz). All six cores are used in
645 the FEM-DEM simulation, while only a single core is used in the FEM-ML simulation.

646

647 Table 4 shows that per iteration in the FEM-ML framework is nearly 82 times faster than the
648 FEM-DEM. In addition to the substantial improvement in efficiency, huge computer memory is no
649 longer needed, thus significantly alleviating the memory requirement of the FEM-DEM multiscale
650 computation and also saving the communicating cost for distributed memory parallel platforms.

651 The number of iterations required at each load step is also recorded in Figure 13 for further
652 comparison. Note that the medium mesh with 4×8 elements, as shown in Figure 6, is also
653 simulated. As shown in Figure 13, both FEM-DEM and FEM-ML methods can converge at every

654 loading step, but more quickly during the initial (near) elastic stage. Then the number of iterations
 655 per loading step significantly increases until the peak is reached at around step 20. Afterwards, the
 656 iterations per step decrease steadily towards around 20 iterations.

657

658 Table 4: Summary of the number of iterations and clock time consumed in the
 659 simulation with the fine-meshed model

	FEM-DEM (multi-cores)	FEM-ML (single)	Speed up
Time (h)	8.02	0.11	72.9
Total iterations	2510	2820	0.89
Time per iteration (s)	11.50	0.14	82.1

660

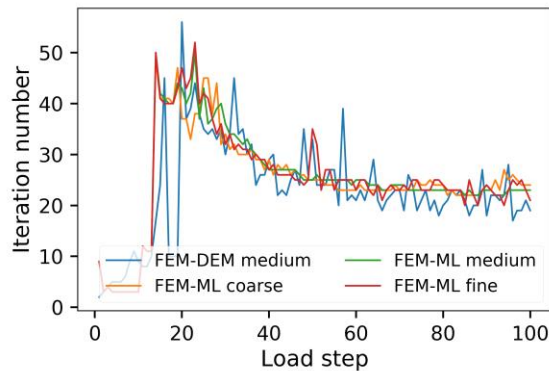
661

662 Compared to the FEM-DEM simulation for the medium mesh, a slightly larger number of
 663 iterations are required by the FEM-ML simulation with all three meshes. This results from the
 664 error between the ML model prediction and the lower-scale RVE simulation. Because of this error,
 665 the FEM-ML framework will in general not converge as fast as the FEM-DEM, but still can obtain
 666 the final equilibrium state through iterations.

667

668 In summary, the results demonstrate that no notable difference is found in the number of iterations
 669 between FEM-ML and FEM-DEM simulations, which further supports the claim that ML models
 670 can be a satisfactory surrogate model for lower-scale RVE simulations.

671

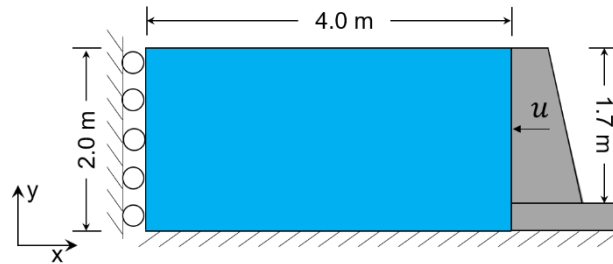


672

673 Figure 13: The number of iterations required for the FEM-DEM and FEM-ML
 674 simulations with the three meshes

675 4.2 Retaining wall example

676 To evaluate the generality of the proposed neural network model, the well-trained neural network
677 model in the previous biaxial compression case is employed in a retaining wall problem. The
678 details of the problem are shown in Figure 14, where the normal constraint is applied to the left
679 boundary; the bottom is constrained in the x and y directions; and a prescribed displacement is
680 applied to the right boundary, acting as the retaining wall, to compress the soil in the normal
681 direction.



682

683

Figure 14: Schematic of the retaining wall problem

684

685 Figure 15a shows a prominent cambered shearing band emerging from the FEM-DEM simulation.
686 The total force applied by the retaining wall versus the transverse strain is depicted in Figure 16.

687

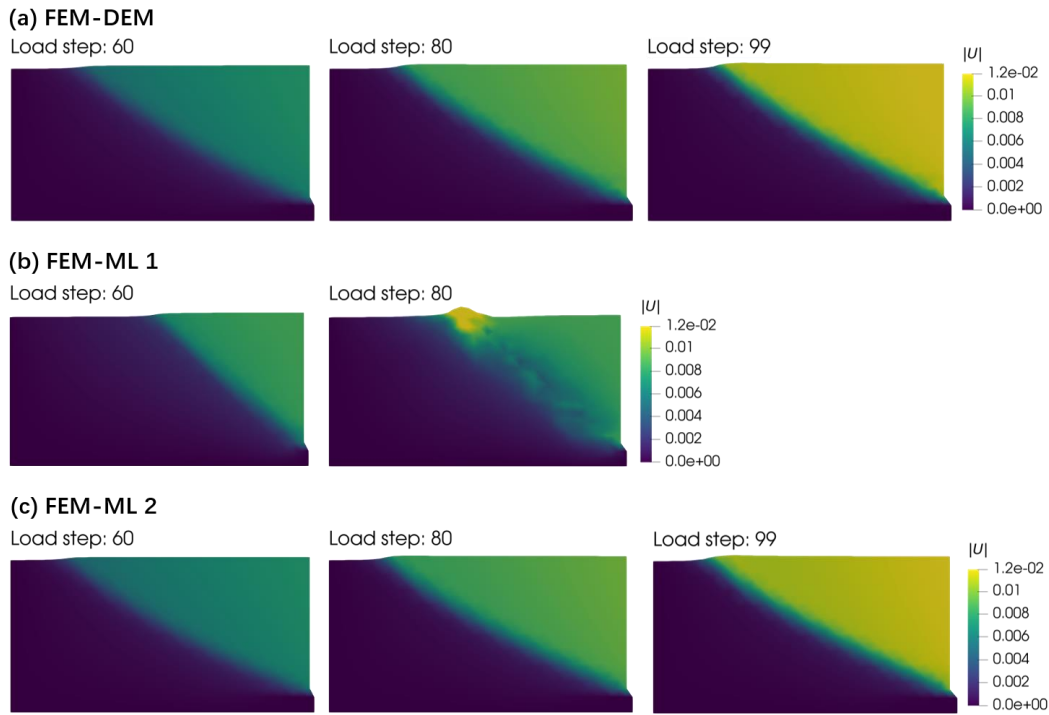
688 Two FEM-ML approaches have been considered. Note that the first FEM-ML approach, labelled
689 *FEM-ML 1*, is trained only on the datasets collected from the biaxial simulations described in
690 Section 3, while the second approach, labelled *FEM-ML 2*, is trained based on the enhanced
691 datasets, as will be explained below.

692

693 In the *FEM-ML 1* simulation, the solution process breaks down at about the 80th load step. The
694 problem arises due to the accumulated error of the strain and the internal variable when their
695 values are far beyond the network training ranges. The shearing band of the *FEM-ML 1* simulation
696 is approximately lying on a straight line, which largely results from the training dataset gathered
697 from the multiscale biaxial simulations, whose shearing band is a straight line. The failure of the
698 *FEM-ML 1* simulation indicates that the neural network model trained based on the data from the
699 biaxial compression test cannot fully reproduce the micromechanical response in the retaining
700 wall simulation due to the limitation of the loading paths in the training samples.

701

702 The *FEM-ML 2* approach uses an enhanced network, where the network used in *FEM-ML 1* is
703 retrained after the datasets of the retaining wall simulations with FEM-DEM are added to the
704 original training samples. Figure 15c and Figure 16 show that the performance of the enhanced
705 network is significantly improved in both displacement and force calculations. Therefore, the
706 proposed methodology is adaptable to upgrade the network model once new datasets are available.

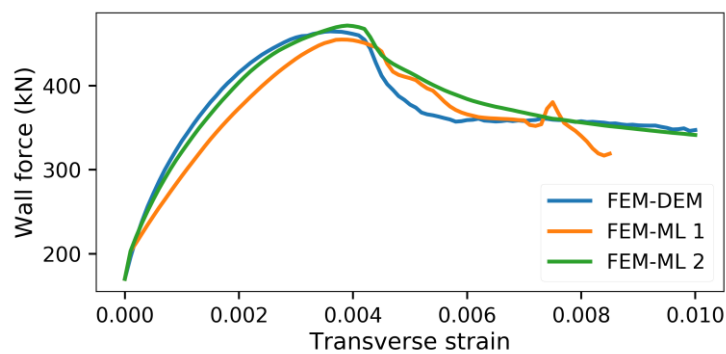


707

708 Figure 15: Displacement distributions of the soil at some load steps when compressed
709 by the retaining wall and obtained by (a) FEM-DEM, (b) FEM-ML 1, and the (c)

FEM-ML 2

711



712

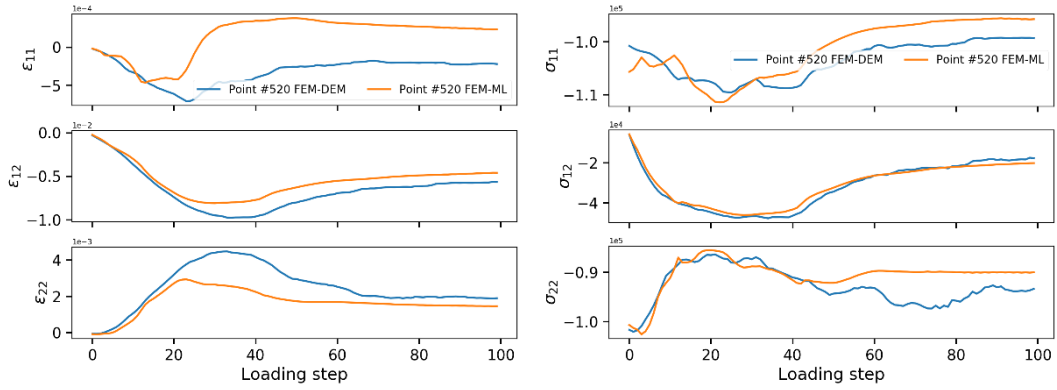
713

Figure 16: The integrated force on the retaining wall

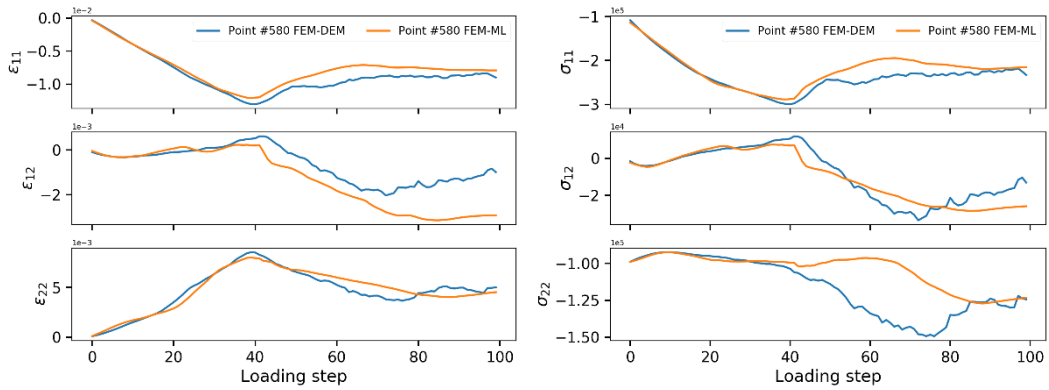
714

We have extracted the strain at the Gaussian point of the retaining wall simulation to illustrate that,

715 as is shown in following figure, unloading has occurred in our example, or that the direction of
 716 strain increment has reversed rather than being a single loading. Without the use of internal
 717 variables to calibrate the granular material state, the network would not be able to reproduce this
 718 path-dependent property.



(a) Points #520 in retaining wall simulation



(b) Points #580 in retaining wall simulation

Figure 17 Curves of stress and strain of Gauss points

5 Conclusion

Our work is primarily devoted to developing a FEM-ML framework and training a network-based constitutive model to replace the micro RVE model via DEM, thus accelerating the classical multiscale FEM-DEM simulation. A multi-layer fully connected neural network, together with the use of the accumulated absolute values of the strain increments as an explicit parametrisation of the strain-stress relationship, is chosen as the surrogate model. This simple network appears to be reasonably able to reproduce the history-dependent mechanical response of granular materials. A Gaussian process-controlled random loading is applied to a biaxial compression problem to

733 generate training samples. In particular, an uncertainty-level-based active learning scheme is
734 utilised to evaluate the informativeness of data points and select the points with high uncertainty
735 levels to enrich the training dataset. This resampling strategy is generic and proved to be highly
736 effective at least for the current problem concerned. A simpler but problem-specific shear strain-
737 based resampling scheme is also discussed.

738

739 The drained biaxial compression tests conducted demonstrate that the FEM-ML framework can
740 genuinely reproduce the micro-scale response of the granular material at a considerably lower
741 CPU cost than the FEM-DEM approach. The generality or extrapolation capability of the
742 proposed framework is also examined in the retaining wall example. The numerical result
743 illustrates that this framework is sufficiently flexible to improve its performance as long as the
744 training datasets are abundant or can be enriched.

745

746 The numerical examples provided highlight that a considerable improvement in the computational
747 efficiency can be made based on the trained surrogate network models which may help to extend
748 the multiscale computational framework to practical engineering problems. With the development
749 of advanced numerical simulations and physical experiments, more high-fidelity datasets will be
750 available. This tendency will contribute to developing a more accurate and general network-based
751 constitutive mode and therefore further promote the application of machine learning-based
752 constitutive models.

753

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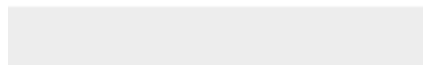
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retaining_gauss_points.rar



Response to the reviewers:

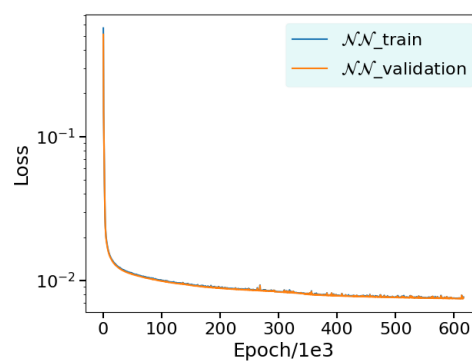
Reviewer #1: The following reviewer comments which were previously mentioned have not been resolved yet, and the authors are encouraged to address them:

We are sorry that we did not reply and express the last revision clearly. Thank you for your interest in our work and for your valuable comments both last time and this time. One important point is the loading/unloading process. The unloading points surely exist in the retaining wall simulation and even in the biaxial compression. We have taken all your suggestions into consideration and thank you for your advice.

1) The number of epochs used and training and test MSE or evolution of the loss function with epochs for test and training datasets need to be provided to demonstrate proper training of the network.

Here is the evaluation of the training loss and the validation loss with the epochs of the network used to predict the datapoints in figure 5. Properly training the network is a very crucial condition for us to complete the ML-based FEM calculation framework. Actually, Figure 5 is intended to show the validity of the network training.

By the way, 'validation' in following figure means the datapoints are split from the training datasets, and never be cast into the training process, only used to make sure the network is not overfitted as is mentioned in Line 450-455. While, points in Figure 5 in the manuscript are totally collected from the test set.



Review2 figure 1 Evolution of the loss with training epochs of the network used in Figure

5

2) The manuscript needs further proofreading. For example, "neutron" has been used instead of "neuron".

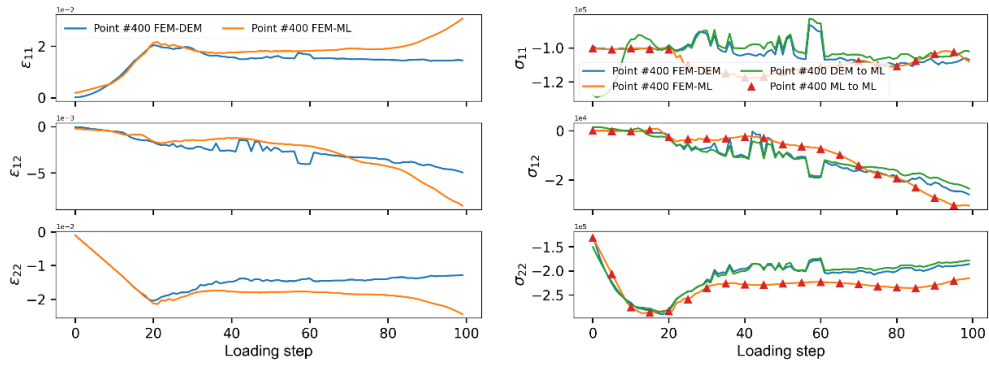
We further proofread the sentences and the logical order of this text.

3) Performance of the model for the case of loading and unloading (e.g. cyclic load) has not been studied in the paper. Therefore, the authors are encouraged to be more precise in their discussion in the paper and explain the extent to which the capability of the model in capturing path-dependent behavior has been studied and potential limitations in capturing path-dependent behavior which are left for future work.

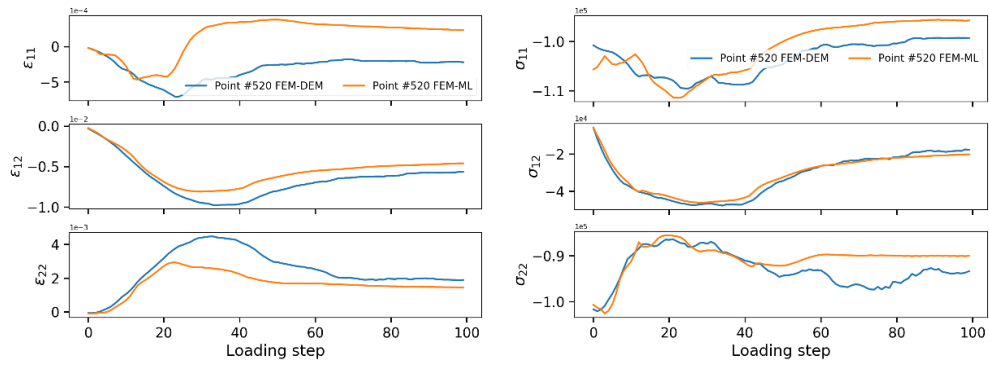
Thanks for your interest in this work. We have introduced the accumulation of the absolute value of strain increment as internal variables to calibrate the state of the granular material. This approach, although not perfect, enables the neural network to describe the path dependent nature of the granular material.

Your concern may be that the biaxial compression example presented in our paper without the add/drop cycle does not reflect the path dependence we mentioned. We have therefore extracted the strain at the Gaussian point of the both simulations to illustrate that, as is shown in following figure, unloading has occurred in our example, or that the direction of strain increment has reversed rather than being a single loading. Without the use of internal variables to calibrate the granular material state, the network would not be able to reproduce this path-dependent property.

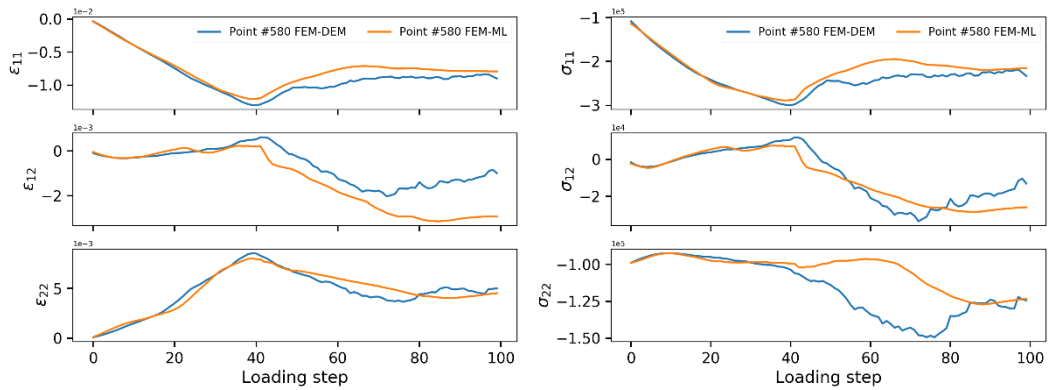
As you can see, we are up to the 100th load step in our loading. It may be necessary to add a few dozen more load steps if we are to finish unloading. A considerable problem with neural networks embedded in BVP calculations is error accumulation because, like recurrent neural networks, the predictions from the previous step are used to calculate the input values for the next prediction. Since our training range is not wide enough, the accuracy of the network drops dramatically when the input values are deviated from the training range. So if longer loading is to be accomplished, there needs to be a corresponding method to reduce the prediction error of the network or slow down the rate of error accumulating. We are still experimenting with this aspect of the work, which is a real challenge for us.



(a) Points #400 in biaxial compression



(a) Points #520 in retaining wall simulation



(b) Points #580 in retaining wall simulation

Review2 figure 2 Curves of stress and strain of Gauss points

4) The authors state in response to Comment 14 that "the network used in FEM-ML 1 is re-trained after the datasets of the retaining wall simulations with FEM-DEM are added to the original training samples". How many samples were added to the training set and what is the size of the final training set?

Also, are the simulation results of the retaining wall BVP in Section 4.2 used for both training (i.e. FEM ML 2 model) and validating the model (i.e. Figure 15(c) and Figure 16)? If so, the outcomes are not valid because a trained network typically performs well on the dataset on which it was trained, thus it cannot be validated using the same dataset used in training.

In the FEM-DEM retaining wall calculations in Section 4.2, we generated close to 2 million additional pairs of data. These data is added to whole the data described in Table 1, from which 0.1 was selected completely at random for training, due to the limited memory of our graphics card and the large number of duplicate samples in the sample. In the end, a total of 1,879,612 pairs of data were used in the training of the FEM-ML 2 computational model.

In fact, we do use all the data from the FEM-DEM retaining wall simulation. Initially we hope that the randomly generated training samples using the Gaussian process described in Section 3.4 would be sufficient to cover the prediction space required for the retaining wall calculations. However, due to the characteristic that granular materials are not able to withstand tensile stresses, it was not possible to design a completely random loading path. FEM-ML 1 did not use the FEM-DEM retaining wall simulation data at all, and thus is not able fully complete the retaining wall calculation, indicating that the sample space is not sufficient. The results of FEM-ML 2 can only be obtained after inputting the FEM-DEM retaining walls datasets.

In response to the question "If so, the outcomes are not valid because a trained network typically performs well on the dataset on which it was trained, thus it cannot be validated using the same dataset used in training.". We have attached plots of the stress-strain curves at Gaussian points calculated in the FEM-ML 2 retaining wall simulation. It can be seen that even when all the data points are put into the network training, the network cannot fully reproduce the results in the FEM-DEM dataset when used again for BVP calculations, and there are even many points where the error at the granular ensemble level is still significant. So it should be a relatively difficult job to reproduce the multi-scale calculation via network calculation even if we have the noisy datasets, especially to calculate at all of the Gauss points and achieve global balance.

Actually, the network's ability to generalise is limited and is only reflected in the prediction of the interpolation range. Once the input exceeds the trained range, the results obtained will be meaningless which is suggested by the retaining wall calculation.

Therefore, this work is used to demonstrate the usefulness of neural networks for improving the

efficiency of multi-scale computation and the challenges of generalisation that a purely data-driven approach will face in BVP computation.

The reviewer has a few additional comments as listed below:

1) Figure 8 shows the simulation results with different coarse and fine meshes from Figure 6. What are the meshes used in the simulations and the presented results?

Yes, the model in Figure 8 is shown in Figure 6. All of the computational details for Figure 8 are taken from Table 3, where we detail the level of grid division for the computational model and what level of grid the data used for network training came from.

2) How are the number of training samples summarized in Table 1 obtained? It seems that 10 loading paths were generated, each with 100 time steps, and were applied as confining pressure of the biaxial BVP in Figure 6. For example, for the coarse mesh, it seems to the reviewer that there are 32 Gauss points, each undergoing 10 loading cases with 100 steps each, leading to 32000 samples.

Sorry we didn't explain the data generation process clearly.

Due to the nonlinear macroscopic mechanical properties of granular materials, each loading step needs nonlinear iteration. On average, a loading step takes about 20 iterations. Therefore, more than 2000 calculation results are generated in 100 loading steps, as is shown in Figure 13. For a coarse mesh, with 32 Gaussian integration points, a single simulation can produce more than $32 \times 100 \times 20 = 64000$ pairs of data.

Reviewer #2: This study is innovative in machine learning use for multiscale computation framework for granular materials. It is interesting and practical. The revision is of considerable quality for publication in AG. It can be accepted in current version.

Thanks for your kind suggestion.